

Magnetic field induced modification of a first-order ferromagnetic transition in Eu_2In

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We present a comprehensive study of the temperature- and magnetic-field-dependent magnetization, specific heat, and local crystal structure across the first-order ferromagnetic–paramagnetic transition in Eu_2In . Anomalies in the magnetocaloric response are observed near $H \approx 25$ kOe, including changes in field scaling of magnetic entropy, local entropy exponent, and universal master curve, which suggest an apparent weakening of the first-order character of the transition. However, quantitative analysis of the magnetocaloric parameters together with modified Arrott plots demonstrates that the transition remains first order up to at least 70 kOe. Specific-heat measurements reveal a field-induced splitting of the sharp zero-field anomaly into a doublet, providing a natural explanation for the change in the magnetocaloric response. Magnetic-field-dependent extended x-ray absorption fine structure (EXAFS) measurements show no detectable field-induced changes in the local coordination environment of Eu. We therefore attribute these observations to a magnetic-field-induced two-step transition process in Eu_2In .

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I. INTRODUCTION

A fundamental understanding of the thermodynamic nature of phase transitions is essential for elucidating the nontrivial responses of materials to applied magnetic fields, which underpin a wide range of magnetofunctional properties [1,2]. In particular, first-order magnetic phase transitions (FOMPTs), characterized by their discontinuous nature, often exhibit strong nonlinear responses to external stimuli and are therefore of considerable interest for functional applications [3–5]. Giant magnetocaloric effects (GMCE), for example, are typically observed in systems undergoing discontinuous magnetic or magnetostructural transitions [5–8]. A prototypical case is $\text{Gd}_5(\text{Si}_2\text{Ge}_2)$, which exhibits a pronounced GMCE near room temperature associated with a first-order phase transition (FOPT) [5,9,10]. In many materials, such transitions originate from strong coupling between magnetic order and lattice degrees of freedom that are inherently accompanied by thermal hysteresis. Although extensive efforts over the past decades have focused on understanding the thermodynamic processes occurring in the vicinity of FOPTs [11–15], hysteresis associated with the transitions remains a major challenge, as it leads to energy dissipation, reduces refrigeration efficiency, and limits long-term cycling stability in solid-state caloric applications [16,17].

In this context, R_2In (R = rare earth) compounds have recently attracted particular attention [18], as certain members of this family exhibit large magnetic entropy changes (ΔS_M) under modest magnetic fields, negligible thermal hysteresis,

and minimal structural distortion at the first-order ferromagnetic (FM) to paramagnetic (PM) transition—characteristics highly desirable for practical magnetocaloric applications [19–25]. Within the R_2In family, compounds with $\text{R} = \text{Eu}$ and Pr exhibit first-order transitions [19,20], $\text{R} = \text{Nd}$ displays a borderline behavior between first- and second-order transitions [21–23], while the remaining members undergo either second-order or no magnetic transitions [18,25,26]. R_2In compounds crystallize in a Co_2Si -type orthorhombic structure for $\text{R} = \text{Eu}$ and Yb [19,27], whereas all other members adopt the Ni_2In -type hexagonal structure [18,28], indicating no direct connection between the crystal structure and the order of the magnetic transition in these compounds. However, the $\text{R} = \text{Eu}$ and Yb members host divalent rare-earth ions [27,29,30], whereas in all other R_2In compounds the rare-earth elements occur in the trivalent state, suggesting an intricate correlation between the valence state and the crystal structure of these compounds.

In this family, Eu_2In exhibits the highest magnetic entropy change of ≈ -28 J/kg K for $\Delta H = 20$ kOe, with negligible thermal hysteresis (0.1 K) and a minimal change of $\approx 0.1\%$ in the unit cell volume across its first-order FM–PM transition at $T_c = 55$ K [19]. Theoretical calculations demonstrate that the strong hybridization between $\text{Eu-}5d$ and $\text{In-}5p$ states near the Fermi level gives rise to long-range exchange interactions between Eu atoms, thereby triggering the ferromagnetic transition, which in turn underlies the large magnetic entropy change at low magnetic fields in this compound [19]. *Ab initio* calculations by Tapia *et al.* attribute the discontinuous phase transition and the associated large latent heat in Eu_2In to a purely electronic mechanism arising from the divalent nature of the Eu ions in the compound [31]. Subsequently,

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Alho *et al.* employed a model Hamiltonian—independent of the details of the electronic structure—that incorporates magnetic exchange and magnetoelastic interaction terms, and successfully reproduces the magnetic-field dependence of T_c as well as the temperature dependence of the magnetization and magnetocaloric properties [32]. Furthermore, by analyzing the evolution of mathematical critical points in the magnetic free energy as a function of magnetization with respect to temperature and applied field, the authors identified a critical magnetic field $H_c \approx 25$ kOe, above which the transition in Eu_2In thermodynamically changes from first order to second order [32]. This theoretically predicted change in the order of the magnetic transition is of particular relevance for magnetocaloric applications, since a reduction in the discontinuity and sharpness of the transition at higher fields can markedly influence the magnetocaloric performance. Nevertheless, no experimental investigations have yet been conducted to verify this proposed field-induced crossover in Eu_2In .

In this study, we report a detailed investigation of the temperature- and field-dependent magnetization, specific heat, and local crystal structure across the first-order FM–PM transition in Eu_2In . Distinct anomalies are identified in the magnetocaloric response, including the maximum magnetic entropy change (ΔS_M^{max}), the local power exponent (n), and the universal scaling behavior, around $H \sim 25$ kOe—the crossover field predicted in Ref. [32]. However, a quantitative analysis of the extracted parameters, together with the modified Arrott plot analysis, demonstrates that the transition retains its first-order nature up to the highest applied field of 70 kOe. The specific heat measurements reveal that the sharp zero-field anomaly evolves into a well-defined doublet with increasing magnetic field, while extended x-ray absorption fine structure (EXAFS) measurements show no detectable field-induced modifications in the local coordination environment within the experimental resolution. This behavior is discussed in terms of the possible emergence of a field-driven two-step process associated with the transition.

II. EXPERIMENT

Polycrystalline Eu_2In was synthesized from stoichiometric amounts of high-purity indium (99.9995%, Alfa Aesar) and europium metal (99.995%, Materials Preparation Center, Ames National Laboratory). The elements (total mass ~ 3 g) were loaded into a tantalum (Ta) crucible and sealed by arc welding under an ultrapure argon atmosphere. Prior to sample loading, the Ta crucible was degassed under high vacuum ($\sim 10^{-6}$ Torr) at 1800 °C for 2 h. The Ta crucible was then sealed in a quartz tube under high-purity He gas to prevent oxidation of the crucible at elevated temperatures. The sample was heated to 900 °C for 12 h in a resistive furnace, flipped, and reheated three times to improve homogeneity, and then annealed at 650 °C for 24 h. After furnace cooling, the sample was extracted from the crucible and handled in an Ar-filled glovebox due to its high reactivity with air.

Room-temperature powder x-ray diffraction (XRD) data were collected using a Rigaku diffractometer with Mo K_α radiation [33]. For the XRD measurements, the sample was ground using a mortar and pestle and sieved through a 36 μm

mesh inside an Ar-filled glovebox. The powdered sample was mounted in a shallow circular recess on a flat glass holder and sealed between two Kapton tapes inside the glovebox. The Rietveld refinement of the XRD data was performed using the FULLPROF suite, with a pseudo-Voigt peak shape and linear interpolation between background points.

Temperature- and field-dependent magnetization measurements were performed using a superconducting quantum interference device (SQUID) magnetometer (MPMS XL-7, Quantum Design, USA). The sample was loaded into an airtight holder inside the glovebox to prevent air exposure. Magnetization isotherms at successive temperatures were recorded by increasing the magnetic field from 0 to 70 kOe in a stable (no-overshoot) mode, followed by decreasing the field from 70 kOe to 500 Oe linearly and from 500 Oe to 0 Oe in an oscillatory mode before proceeding to the next temperature, taking into account the low coercivity of the sample (~ 50 Oe). Specific-heat measurements were carried out using a Physical Property Measurement System (PPMS, Quantum Design, Inc.) employing both the standard 2τ relaxation technique and the long heat-pulse method [34,35].

Temperature- and field-dependent x-ray absorption spectroscopy (XAS) measurements at the Eu L_3 edge were performed at the PIPOXS beamline (ID2A) of the Cornell High Energy Synchrotron Source (CHESS) using the NJIT 10 T superconducting Oxford Spectromag4000-10 magnet system. Eu_2In powder was ground and homogenized with boron nitride (BN) in a fixed ratio and pressed into a 13-mm-diameter pellet inside an inert-atmosphere glovebox. The pellet was sealed between two Kapton tapes to prevent air exposure. The sample was cooled in He vapor. At each temperature and magnetic field, four spectra were collected and averaged. Energy calibration was performed using a simultaneously measured Eu metal foil reference to correct for possible energy shifts between scans [36,37]. X-ray absorption near-edge structure (XANES) and EXAFS data reduction and analysis were performed using ATHENA and ARTEMIS software packages (DEMETEER suite) [38]. Background subtraction and normalization were carried out in ATHENA and EXAFS fitting was performed in ARTEMIS using theoretical scattering paths calculated with FEFF6 [39].

III. RESULTS

A. X-ray diffraction

The room-temperature powder XRD pattern of Eu_2In is shown in Fig. 1(a). Rietveld refinement confirms that Eu_2In crystallizes in the Co_2Si -type orthorhombic structure (space group $Pnma$), in agreement with previous reports [19,29]. However, careful inspection reveals a small amount ($\sim 2\%$) of a secondary Eu_8In_3 phase, indicated by an asterisk in Fig. 1(a) (not included in the refinement). This trace amount of the secondary phase in Eu_2In is not uncommon and has also been observed previously [27]. In the present study we focus primarily on field-induced changes in the nature of the transition in Eu_2In around $T_c = 55$ K; therefore, this minor secondary phase is not expected to influence the conclusions of our findings. The increasing background at low 2θ values ($\lesssim 20^\circ$) arises from the Kapton tape used to protect the sample from

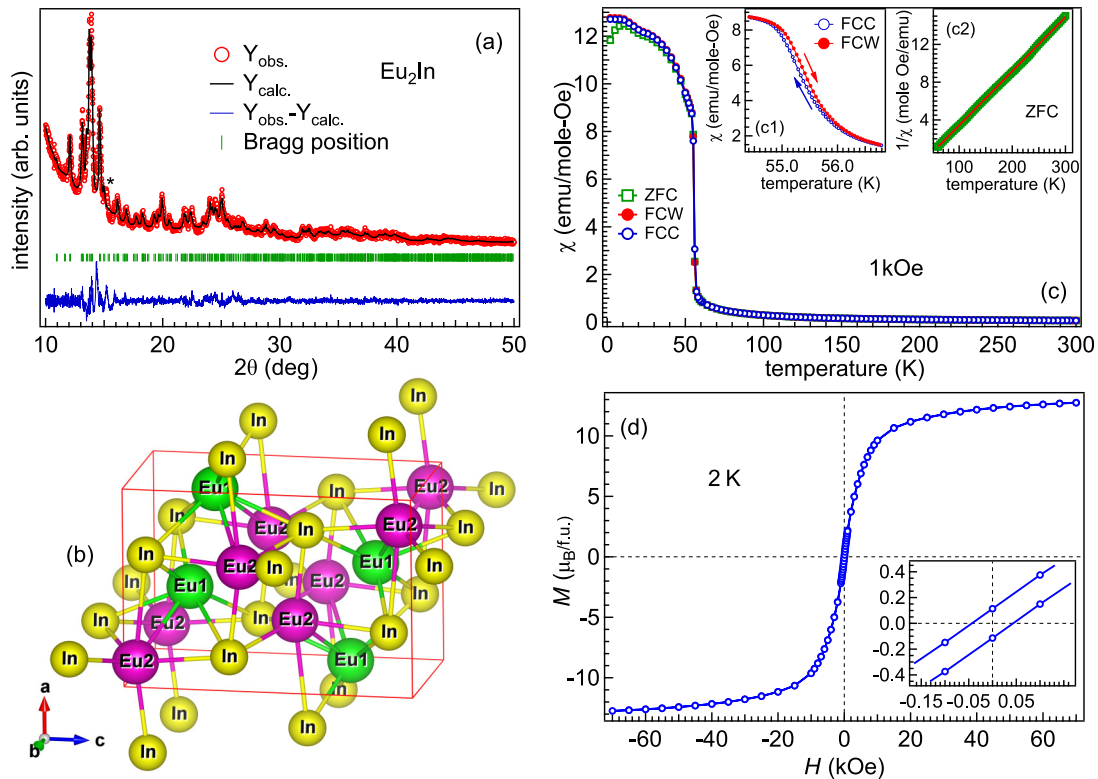


FIG. 1. (a) Rietveld refinement of the room-temperature powder XRD pattern of Eu_2In collected using $\text{Mo } K_\alpha$ radiation. Red symbols represent the observed intensities and the black solid line denotes the calculated profile. Vertical tick marks indicate the Bragg positions corresponding to the orthorhombic $Pnma$ (Co_2Si -type) phase. The bottom trace shows the difference between the observed and calculated patterns. An asterisk indicates the presence of a small amount ($\sim 2\%$) of a secondary Eu_8In_3 phase. (b) Three-dimensional representation of the crystal structure of Eu_2In , highlighting the two inequivalent Eu sites. (c) Temperature-dependent magnetic susceptibility (χ - T) measured in zero field cooled (ZFC), field cooled warming (FCW), and field cooled cooling (FCC) protocols at 1 kOe in the temperature stable mode. Inset (c1) shows high resolution FCC and FCW data recorded in the temperature sweep mode in the vicinity of T_c . Inset (c2) shows the Curie-Weiss fit to the FCW data between 60 and 300 K. (d) Field-dependent magnetization measured at 2 K. The inset shows an enlarged view of the low-field region.

air exposure. The refined lattice parameters, atomic Wyckoff positions, and refinement quality indicators are summarized in Table I. Figure 1(b) shows a three-dimensional representation of the Eu_2In crystal structure, highlighting the two crystallographically inequivalent Eu sites.

B. Magnetization

The temperature-dependent magnetic susceptibility, $\chi = M/H$, of Eu_2In measured at an applied field of $H = 1$ kOe under zero field cooled (ZFC), field cooled warming (FCW), and field cooled cooling (FCC) protocols is shown in Fig. 1(c). A sharp magnetic transition with no apparent thermal

hysteresis is observed at $T_c \approx 55$ K, consistent with the previously reported first-order ferromagnetic to paramagnetic transition [19]. However, the “temperature settle” mode of the magnetization measurement is not suitable for resolving the small thermal hysteresis associated with a first-order transition. This limitation arises because even a slight overshoot of the target temperature can drive the system across the phase boundary, transforming it from one phase to the other; upon returning to the set temperature, the system may remain trapped in the transformed state. Therefore, to reliably capture the hysteretic behavior near T_c , magnetization measurements were performed upon both heating and cooling using the “temperature sweep” mode with a sweep rate of 0.1 K/min,

TABLE I. Room-temperature structural parameters and refinement quality indicators of Eu_2In obtained from Rietveld refinement of powder XRD data collected using $\text{Mo } K_\alpha$ radiation. Values in parentheses represent the standard deviations of the last significant digits.

Lattice parameters		Atomic position	Site	x	y	z	Fitting parameters
a (Å)	7.4507(7)	Eu1	4c	0.0322(10)	0.25	0.7074(6)	$\chi^2 = 1.96$
b (Å)	5.5795(5)	Eu2	4c	0.1802(7)	0.25	0.0715(6)	$R_p = 5.11$
c (Å)	10.3128(10)	In	4c	0.2380(12)	0.25	0.3926(7)	$R_{wp} = 7.39$
Cell volume (Å ³)	428.72(7)						

as shown in the inset (c1) of Fig. 1(c). A clear thermal hysteresis of ~ 0.1 K is observed at T_c , confirming the first-order nature of the transition. Furthermore, the ZFC susceptibility was fitted to the Curie-Weiss law, $\chi = C/(T - T_\theta)$, where C is the Curie constant and T_θ is the Curie-Weiss temperature. The linear fit of $1/\chi$ versus T over the temperature range 60–300 K, shown in the inset (c2) of Fig. 1(c), yields $T_\theta = 40.1$ K and an effective magnetic moment of $\mu_{\text{eff}} = 8.36\mu_B/\text{Eu}$. The fact that $T_\theta < T_c$ is attributed to the very sharp first-order character of the transition. The extracted μ_{eff} is slightly larger than the free-ion value expected for Eu^{2+} ($7.94\mu_B$), which may indicate a small additional contribution from exchange polarization of itinerant conduction band states (with Eu 5d character) by the localized Eu 4f moments, rather than a purely localized 4f response. Figure 1(d) shows the field dependence of the magnetization measured at 2 K. The magnetization does not fully saturate even up to $H = 70$ kOe. At 70 kOe, the moment reaches $\sim 12.7\mu_B/\text{f.u.}$, which is lower than the value expected for complete ferromagnetic alignment of two Eu^{2+} ions ($gJ = 7\mu_B/\text{Eu}$). The hysteresis loop exhibits a small coercive field of ~ 50 Oe, as highlighted in the inset showing an enlarged view of the low-field region. Overall, Eu_2In combines minimal thermal and magnetic hysteresis with a pronounced change in magnetization at T_c , characteristics that are favorable for magnetocaloric applications [5,19].

To investigate the magnetic-field-induced changes in the nature of this transition in Eu_2In , the virgin magnetization isotherms have been recorded across T_c with a temperature step of $\Delta T = 0.5$ K [see Fig. 10(a) of the Appendix]. The magnetic entropy change (ΔS_M) is then calculated from these isotherms by employing the Maxwell thermodynamic relation [40,41]

$$\Delta S_M(T, H) = \int_0^H \left(\frac{\partial M(T, H)}{\partial T} \right)_H dH. \quad (1)$$

The $\Delta S_M(T)$ curves for Eu_2In are shown in Fig. 2(a) for $\Delta H = 2.5$ –70 kOe, revealing a GMCE with a maximum magnetic entropy change (ΔS_M^{max}) of ~ -20 J/kg K for $\Delta H = 20$ kOe. The value of ΔS_M^{max} increases rapidly with increasing ΔH up to ~ 25 kOe and then rises at a much slower rate at higher fields, as shown in the inset of Fig. 2(a) on a log-log scale. For a second-order FM–PM transition, ΔS_M^{max} is expected to follow a power-law dependence on the applied magnetic field, $\Delta S_M^{\text{max}} \propto H^n$ [42]. In Eu_2In , the $\Delta S_M^{\text{max}}(H)$ curve deviates from this power-law behavior for $H < 25$ kOe, as evidenced by the nonlinearity in the log-log plot presented in the inset of Fig. 2(a) [see Figs. S1(a) and S1(b) of Supplemental Material [43] for more clarity]. In contrast, an almost linear behavior is observed for $H > 25$ kOe, suggesting a qualitative change in the magnetic response above this field. A power-law fit to the 25–70 kOe data (solid black line in the inset) yields $n = 0.24$, which is lower than the theoretical value for any established universality class of FM materials [42]. This suggests that although a notable change in the magnetic behavior of the sample is observed at $H \sim 25$ kOe, as predicted in Ref. [32], the overall nature of the phase transition does not transform into a purely second-order type even for $H > 25$ kOe.

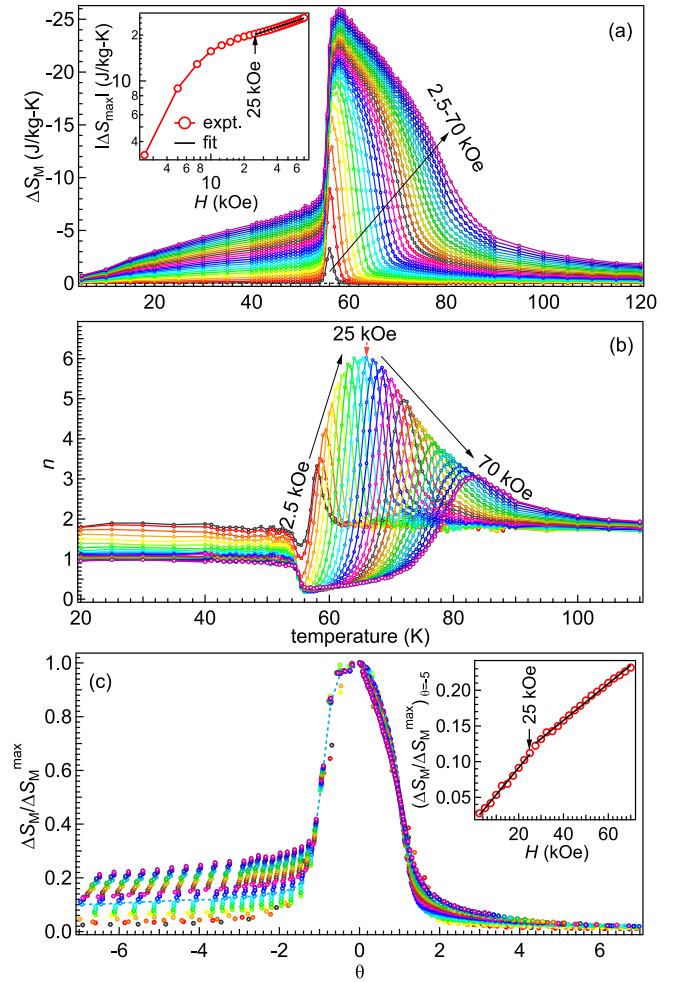


FIG. 2. (a) Temperature-dependent magnetic entropy change (ΔS_M) of Eu_2In at different magnetic fields. The inset shows the field dependence of the maximum magnetic entropy change (ΔS_M^{max}) on a log-log scale, where the solid black line represents the linear fit for $H > 25$ kOe. (b) Temperature-dependent local field exponent (n) of ΔS_M at different fields. (c) Normalized ΔS_M versus scaled temperature curves at different fields, with the dashed line representing the curve for $H = 25$ kOe. The inset shows the field dependence of the vertical dispersion at $\theta = -5$, where the solid black lines represent straight fits to the data below and above $H = 25$ kOe.

Further, the $\Delta S_M(T)$ curves remain nearly symmetric up to $H = 25$ kOe and then gradually become asymmetric towards the PM region at higher fields [see Fig. 2(a)]. To better understand this, we present the local field exponent of entropy, n , defined as [44]

$$n(H, T) = \frac{d(\ln|\Delta S_M|)}{d(\ln H)} \quad (2)$$

in Fig. 2(b) at various magnetic fields. Note that, for a second-order FM–PM transition, $n = 1$ in the FM state, attains a minimum at T_c , given by $n(T_c) = 1 + (1 - \beta^{-1})\delta^{-1}$, where β and δ are the critical exponents, and approaches $n = 2$ in the PM region as expected from Curie-Weiss law [42]. In contrast, $n > 2$ signifies a first-order transition [44]. Interestingly, in Eu_2In , n initially increases monotonically in the vicinity of T_c , attains its maximum value of $n \sim 6$ at $H = 25$ kOe, and

then decreases with further increase in the magnetic field [see Fig. 2(b)]. This suggests a possible transformation from first-order to second-order character of this FM–PM phase transition in Eu_2In at $H = 25$ kOe. Nevertheless, $n > 2$ persists up to 70 kOe, which indicates that the overall nature of the transition remains first order up to 70 kOe, but its first-order character likely decreases above 25 kOe.

To further clarify this, we construct a universal master curve (UMC) of $\Delta S_M(T)$ data measured at different magnetic fields by plotting the normalized entropy ($\Delta S_M/\Delta S_M^{\max}$) versus scaled temperature (θ), defined as

$$\theta = \begin{cases} -(T - T_{pk})/(T_{r1} - T_{pk}), & T \leq T_c, \\ (T - T_{pk})/(T_{r2} - T_{pk}), & T > T_c, \end{cases} \quad (3)$$

where T_{r1} and T_{r2} are the two reference temperatures below and above T_c , respectively, for a certain value of $h = \Delta S_M/\Delta S_M^{\max}$, and T_{pk} represents the temperature corresponding to $\Delta S_M^{\max}$ [42]. The normalized entropy versus scaled temperature for Eu_2In is presented in Fig. 2(c) at different applied magnetic fields for $h = 1/2$. For a second-order FM–PM transition, curves at different fields collapse into a single universal master curve in the entire temperature range, whereas a vertical dispersion is observed for first-order transition below T_c [45]. In the present case, a clear dispersion in the scaled curves is observed as a function of H below T_c [see Fig. 2(c)]. Interestingly, the curves exhibit considerably larger dispersion up to $H \approx 25$ kOe [indicated by the dashed line in Fig. 2(c)], which is subsequently reduced at higher magnetic fields. To clearly demonstrate this, we plot the field dependence of the normalized entropy at $\theta = -5$ in the inset of Fig. 2(c). A clear decrease in the dispersion rate is observed around 25 kOe, where the two solid black lines are straight fits to the data below and above $H = 25$ kOe. However, although at a lower rate, the scaled entropy curves still show dispersion up to 70 kOe, which further indicates only a reduction in the first-order-like character of this magnetic transition above 25 kOe, but not a complete transformation to second-order nature, analogous to the behavior of $\Delta S_M^{\max}$ and n , discussed above.

C. Specific heat

The zero-field specific heat (C_p) of Eu_2In , measured as a function of temperature using the 2τ relaxation method, is shown in Fig. 3(a). A sharp, δ -like peak is clearly observed at $T_c \approx 55$ K, indicating the first-order nature of the magnetic transition. At low temperatures, the specific heat can be expressed as

$$C_p = \gamma T + \beta T^3 + AT^{3/2}, \quad (4)$$

where the first, second, and third terms correspond to electronic, lattice (phonon), and ferromagnetic magnon contributions, respectively [46]. A fit to the C_p data using the above equation in the range ~ 2 –6.5 K, shown by the solid black line in the inset of Fig. 3(a), yields $\gamma = 9.0(7)$ mJ mol $^{-1}$ K $^{-2}$, $\beta = 11.7(5)$ mJ mol $^{-1}$ K $^{-4}$, and $A = 24.1(4)$ mJ mol $^{-1}$ K $^{-5/2}$. Using this value of γ and the Debye model for the lattice heat capacity, we fit the specific heat data in the high-temperature paramagnetic regime (70–200 K) by

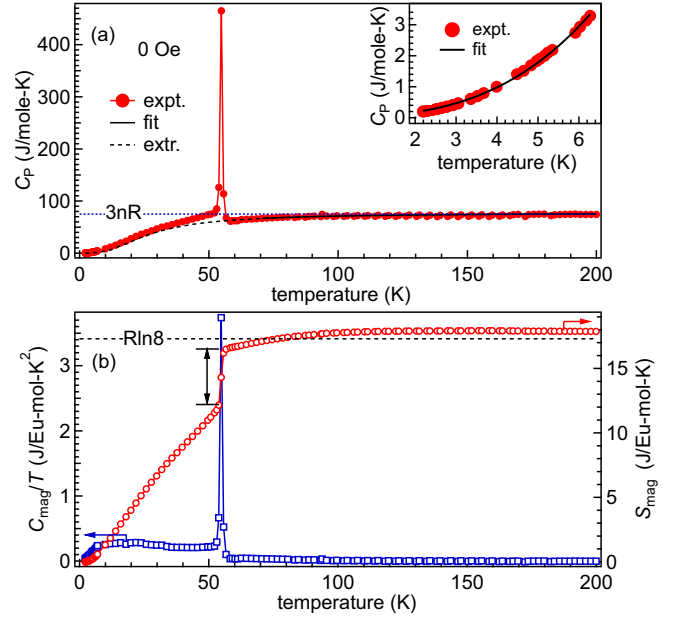


FIG. 3. (a) Temperature-dependent zero-field specific heat of Eu_2In measured using the standard 2τ relaxation method. The solid and dashed black curves represent the best fit of the data using Eq. (5) in the temperature range 70–100 K and the extrapolation of the fitted curve down to 2 K, respectively. The dotted horizontal line represents the classical Dulong-Petit limit of C_p for Eu_2In . The inset shows the enlarged view of the low temperature region, where the solid black line represents the best fit using Eq. (4). (b) Magnetic contribution to the specific heat, C_{mag}/T (left axis), and the magnetic entropy, S_{mag} (right axis), of Eu_2In . The dashed horizontal line represents the theoretical magnetic entropy $S_{\text{mag}} = R \ln(2J + 1)$ expected for Eu^{2+} . The double-sided black arrow indicates the magnetic entropy associated solely with the magnetic transition.

following relation [47,48]:

$$C_p = \gamma T + 9nR \left(\frac{T}{\theta_D} \right)^3 \int_0^{\theta_D/T} \frac{x^4 e^x}{(e^x - 1)^2} dx, \quad (5)$$

where n is the number of atoms per formula unit (three in the present case), R is the universal gas constant, and θ_D is the Debye temperature. The best fit of the C_p data, shown by the solid black curve in Fig. 3(a), gives $\theta_D = 113 \pm 2$ K, which is in good agreement with the value reported for Pr_2In [49]. The dashed curve in Fig. 3(a) represents the extrapolation of the fitted curve down to 2 K, while the dotted horizontal line denotes the classical Dulong-Petit limit, $3nR = 74.8$ J mol $^{-1}$ K $^{-1}$.

The magnetic contribution to the specific heat, C_{mag} , is obtained by subtracting the lattice contribution estimated from the nonmagnetic isostructural reference compound Yb_2In [18], by approximating the lattice heat capacity of Eu_2In to be equal to that of Yb_2In (see Fig. 11 of the Appendix). The resulting C_{mag} is then used to calculate the magnetic entropy, using

$$S_{\text{mag}}(T) = \int_0^T \frac{C_{\text{mag}}(T)}{T} dT. \quad (6)$$

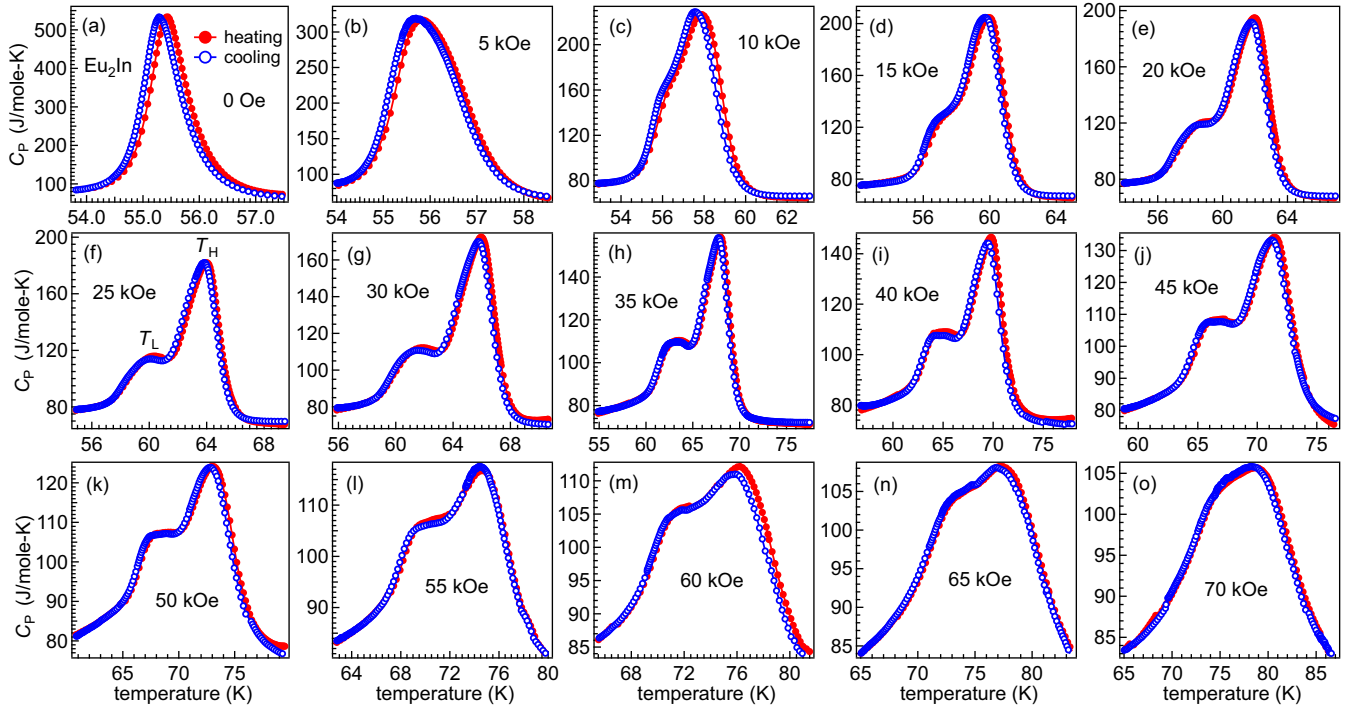


FIG. 4. (a)–(o) Temperature-dependent specific heat of Eu_2In in the vicinity of magnetic transition at different magnetic fields measured in both heating and cooling modes, using the long heat pulse method. T_H and T_L in panel (f) represent the temperatures corresponding to the HT and LT peaks, respectively, at 25 kOe.

The C_{mag}/T and S_{mag} are shown on the left and right axes of Fig. 3(b), respectively. The dashed horizontal line represents the theoretical magnetic entropy value $S_{\text{mag}} = R \ln(2J + 1)$ expected for Eu^{2+} with $J = 7/2$. Interestingly, despite the sharp first-order δ -like anomaly in C_P , only about 25% of the total Eu^{2+} magnetic entropy is released at the transition itself [as indicated by the double-sided arrow in Fig. 3(b)], with the remaining entropy being recovered gradually at lower temperatures. This implies that the magnetic entropy change ΔS_M in Eu_2In could, in principle, be enhanced by approximately a factor of 4 if the full magnetic entropy were utilized, for example, by tuning the free-energy landscape through chemical substitution while preserving the large magnetic moment of the ferromagnetic state [50].

Note that the standard 2τ relaxation method for specific heat measurements assumes that (i) the sample's specific heat remains constant within the small temperature interval raised by a single heat pulse and (ii) the heating and cooling cycles yield identical results for a given pulse [34,35]. However, both the assumptions break down in the case of a sharp (δ -like) first-order transition, leading to inaccuracies in both the position and magnitude of the anomaly [34,35]. Therefore, to accurately capture the first-order transition in Eu_2In , we apply a long heat pulse just below the transition temperature and analyze heating and cooling cycles separately [34,51]. The temperature-dependent specific heat measurements performed using the long heat pulse method under various external magnetic fields are shown in Figs. 4(a)–4(o).

In the absence of an external magnetic field, a sharp peak is observed in the $C_P(T)$ curves with a thermal hysteresis of ~ 0.1 K [Fig. 4(a)]. Interestingly, with increasing magnetic

field, this peak evolves into a well-defined doublet, indicating the emergence of an additional transition. A similar field-induced splitting of the sharp zero-field C_P peak in Eu_2In has also been observed in Ref. [19]. Once the two peaks are sufficiently separated ($H \gtrsim 20$ kOe), the intensity of the low-temperature (LT) peak remains nearly invariant, whereas that of the high-temperature (HT) peak decreases monotonically with further increase in the magnetic field up to 70 kOe [see Fig. 12 of the Appendix]. However, the magnetic entropy

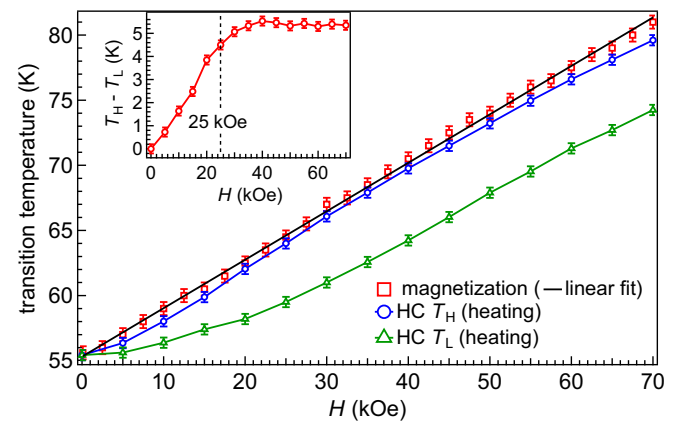


FIG. 5. Magnetic-field dependence of the transition temperatures determined from magnetization and specific-heat measurements. The solid black line represents a linear fit to the $T_c(H)$ curve, extracted from the magnetization data. The inset shows the field-dependent separation between the two peaks observed in the specific heat curves.

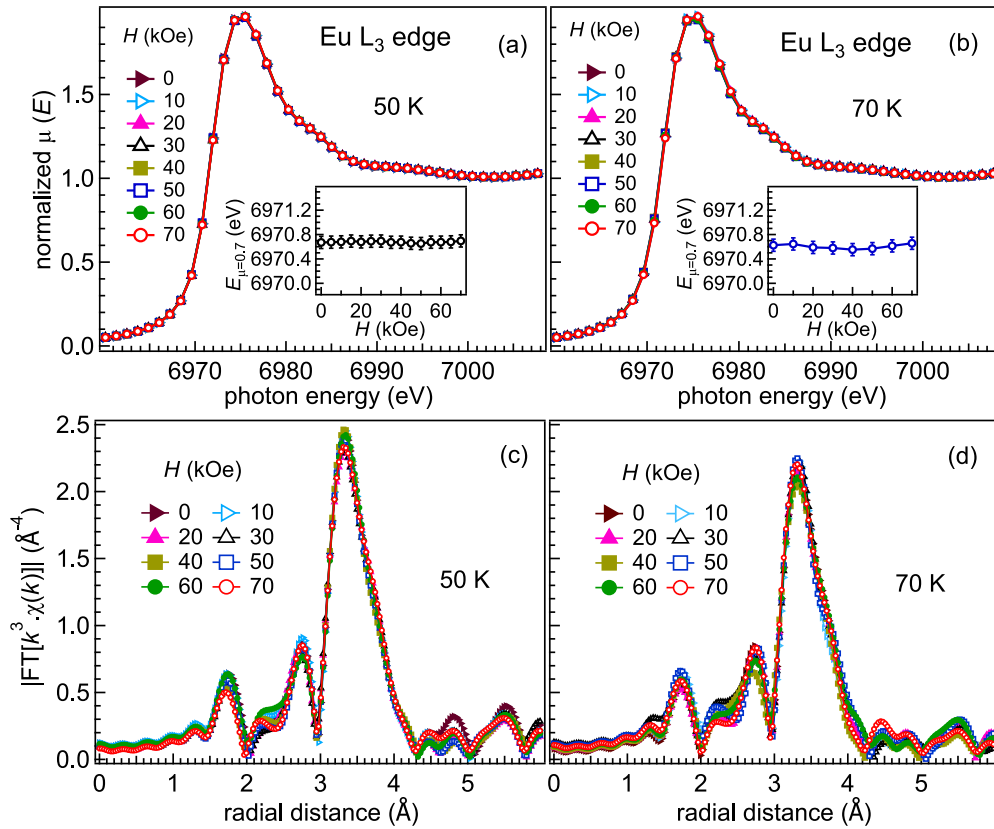


FIG. 6. (a), (b) Normalized XANES spectra at the Eu L_3 edge of Eu₂In measured under different magnetic fields at 50 and 70 K, respectively. The insets show the energy position of the rising edge at $\mu = 0.7$ as a function of magnetic field. (c), (d) Magnitude of the Fourier transform of the $k^3\chi(k)$ EXAFS spectra measured at different magnetic fields at 50 and 70 K, respectively.

associated with the transitions remains nearly invariant with the external field (see the inset of Fig. 13 of the Appendix and the discussion therein). To capture the wide temperature range at higher magnetic fields, where the transition becomes significantly broadened ($H \gtrsim 30$ kOe), more than one heat pulse was employed. A reduction in the thermal hysteresis of the $C_p(T)$ curves is also observed with increasing magnetic field. It should be noted, however, that the true value of the thermal hysteresis cannot be reliably determined at fields where multiple heat pulses are required to cover the transition region.

Here it is important to emphasize that magnetization measurements reveal only a single transition, manifested as a sharp and symmetric peak in the dM/dT curves [Figs. 10(d)–10(f) of the Appendix]. To elucidate the origin of the two anomalies observed in the $C_p(T)$ curves, we compare their peak positions with the transition temperature, T_c , extracted from magnetization measurements [Fig. 10(f) of the Appendix], as shown in Fig. 5. The specific-heat peak positions were determined by fitting the data using a sum of two Voigt functions (Figs. S2(a)–S2(c) of Ref. [43]). Notably, the position of the HT peak (T_H) closely follows T_c obtained from magnetization, indicating its magnetic origin, which is further supported by the systematic reduction of its magnitude with increasing field (Fig. 12 of the Appendix).

In contrast, no corresponding feature is observed in the magnetization data near the LT peak of the $C_p(T)$ curves, indicating that this anomaly is not directly associated with a magnetic phase transition. A natural interpretation is that

the LT anomaly originates from a lattice-related instability that becomes progressively decoupled from the magnetic transition under an applied magnetic field, consistent with the almost field independence of its magnitude (Fig. 12). However, the systematic shift of both the LT and HT peaks to higher temperatures with increasing magnetic field argues against a purely structural transition that is entirely independent of magnetism. Instead, this behavior points to the presence of finite magnetoelastic coupling, whereby the applied magnetic field indirectly influences the lattice degrees of freedom. In this context, a field-induced structural response, such as a magnetostriction-driven effect, remains a plausible scenario.

A key observation is that the separation between the two peaks in the specific-heat curves increases rapidly with magnetic field up to ~ 25 kOe and then remains nearly constant at higher fields, as shown in the inset of Fig. 5, while both transitions broaden with increasing field. This decoupling of the two transitions near $H \sim 25$ kOe may be responsible for the apparent suppression of the first-order-like character of the transition above this critical field, as inferred from the magnetization measurements discussed above. A comparison of the C_p data obtained using the standard 2τ relaxation method and the long heat-pulse method at 35 kOe shows excellent reproducibility (Fig. S3 of Ref. [43]), confirming that the sharp zero-field anomaly indeed splits into two distinct features under applied magnetic field.

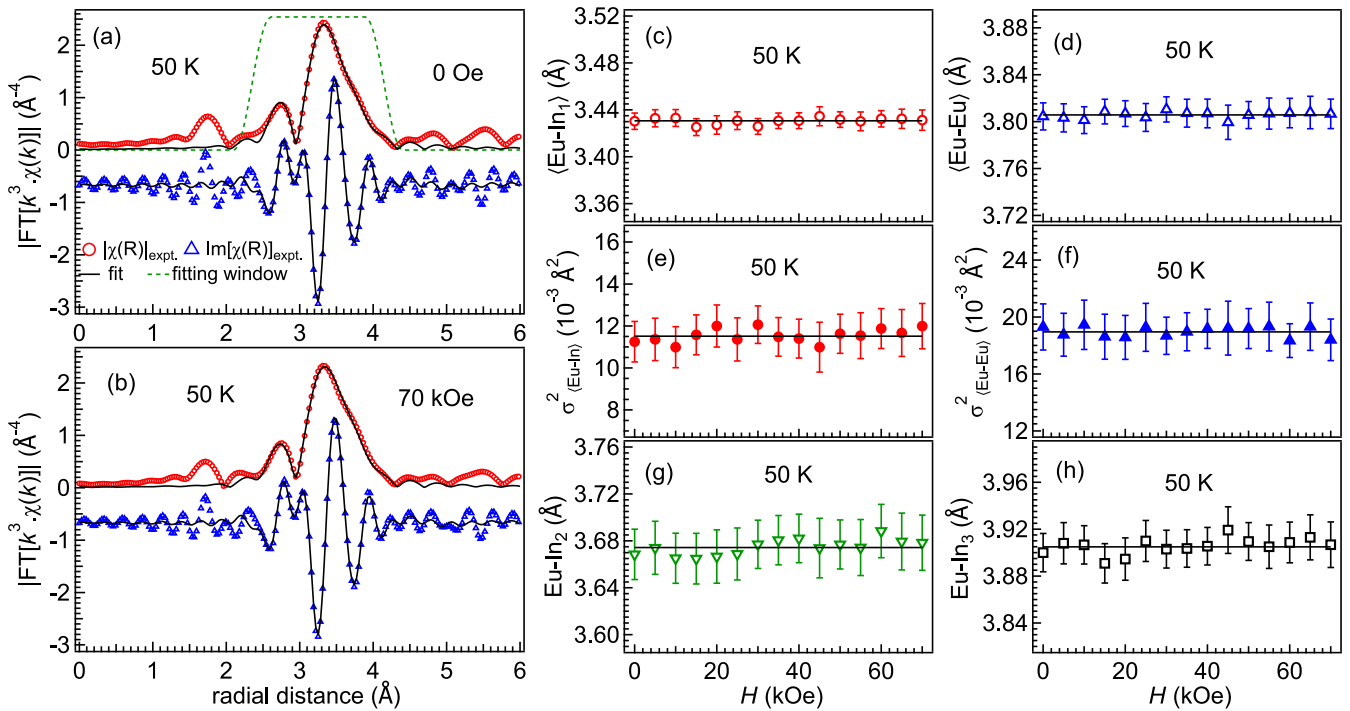


FIG. 7. (a), (b) Fits to the $|\chi(R)|$ and corresponding $\text{Im}[\chi(R)]$ spectra within the window indicated by the dotted green line [in panel (a)] at magnetic fields of 0 and 70 kOe, respectively, at 50 K. Each $\text{Im}[\chi(R)]$ spectrum is shifted downward for clarity. (c), (d) Magnetic-field dependence of the average Eu-In₁ and Eu-Eu scattering path distances, respectively. (e), (f) Corresponding Debye-Waller factors. (g), (h) Magnetic-field dependence of the Eu-In₂ and Eu-In₃ scattering path distances. Solid black lines indicate the field-independent behavior of the various scattering distances and their corresponding Debye-Waller factors.

D. X-ray absorption spectroscopy

Magnetic-field-dependent x-ray absorption spectroscopy (XAS) measurements were performed to probe possible field-induced changes in the local crystal structural and/or electronic environment of Eu₂In. Figures 6(a) and 6(b) show the normalized XANES spectra measured at the Eu L_3 edge under different magnetic fields at 50 K ($<T_c$) and 70 K ($>T_c$), respectively. An intense feature around 6976 eV is attributed to the $2p_{3/2} \rightarrow 5d$ transition of Eu²⁺ [52–54], indicating a predominantly divalent state of Eu in the sample, consistent with the magnetization measurements discussed above and with previous reports [19,30]. More importantly, the valence state of Eu remains invariant with applied magnetic field at both temperatures. Indeed, the entire XANES line shape remains unchanged with H , including the position of the rising-edge region, as illustrated in the insets of Figs. 6(a) and 6(b) at $\mu = 0.7$, where μ denotes the normalized absorption coefficient. These results demonstrate that the electronic structure of Eu₂In is robust against the application of an external magnetic field.

To further investigate the local coordination environment around Eu atoms, we performed EXAFS measurements at the Eu L_3 edge (Figs. S4(a) and S4(b) of Ref. [43]). Figures 6(c) and 6(d) show the magnitude of the Fourier transform of the $k^3\chi(k)$ spectra (Figs. S4(c) and S4(d) of Ref. [43]) measured at different magnetic fields for $T = 50$ and 70 K, respectively. The $\chi(R)$ spectra were obtained using a Hanning window over the k range 2.5–11.5 Å^{−1} with a background cutoff of $R_{\text{bkg}} = 1.8$ Å. The dominant feature at $R \sim 3.5$ Å (phase

uncorrected) arises primarily from nearest-neighbor Eu-In single-scattering paths. Upon closer inspection, this peak exhibits a noticeable asymmetry toward higher radial distance, indicating additional contributions from Eu-Eu correlations and/or longer Eu-In scattering paths. In the orthorhombic $Pnma$ structure, Eu atoms occupy two crystallographically distinct sites (Eu1 and Eu2). Each Eu atom is coordinated by five In atoms (within 4.5 Å), resulting in four distinct Eu-In bond distances. Because the two shortest Eu-In distances (one singly coordinated and one twofold coordinated) are close and cannot be reliably resolved within the present EXAFS resolution, we modeled them using an average distance $\langle\text{Eu}-\text{In}_1\rangle$, together with independent $\langle\text{Eu}-\text{In}_2\rangle$ and $\langle\text{Eu}-\text{In}_3\rangle$ distances, where In_n denotes the n th nearest-neighbor In atom. To avoid overparametrization, a common Debye-Waller (DW) factor was used for all Eu-In paths.

Further, both Eu1 and Eu2 sites are coordinated by eight Eu atoms, comprising five distinct Eu-Eu distances. Since the spread among these Eu-Eu distances is below the resolution of the present EXAFS data ($\lesssim 0.17$ Å), we modeled them using a single average Eu-Eu bond length $\langle\text{Eu}-\text{Eu}\rangle$, with equal weighting of contributions from both Eu sites. The $\chi(R)$ spectra were first fitted at $T = 10$ K in zero field (not shown) and the amplitude reduction factor, S_0^2 , and energy shift, ΔE_0 , were subsequently fixed for all other temperatures and magnetic fields. Fits were performed over the range $R = 2.35$ –4.15 Å [green dashed curves in Figs. 7(a) and 8(a)]. Representative fits at 0 and 70 kOe are shown in Figs. 7(a) and 7(b) for 50 K and in Figs. 8(a) and 8(b) for 70 K, respectively

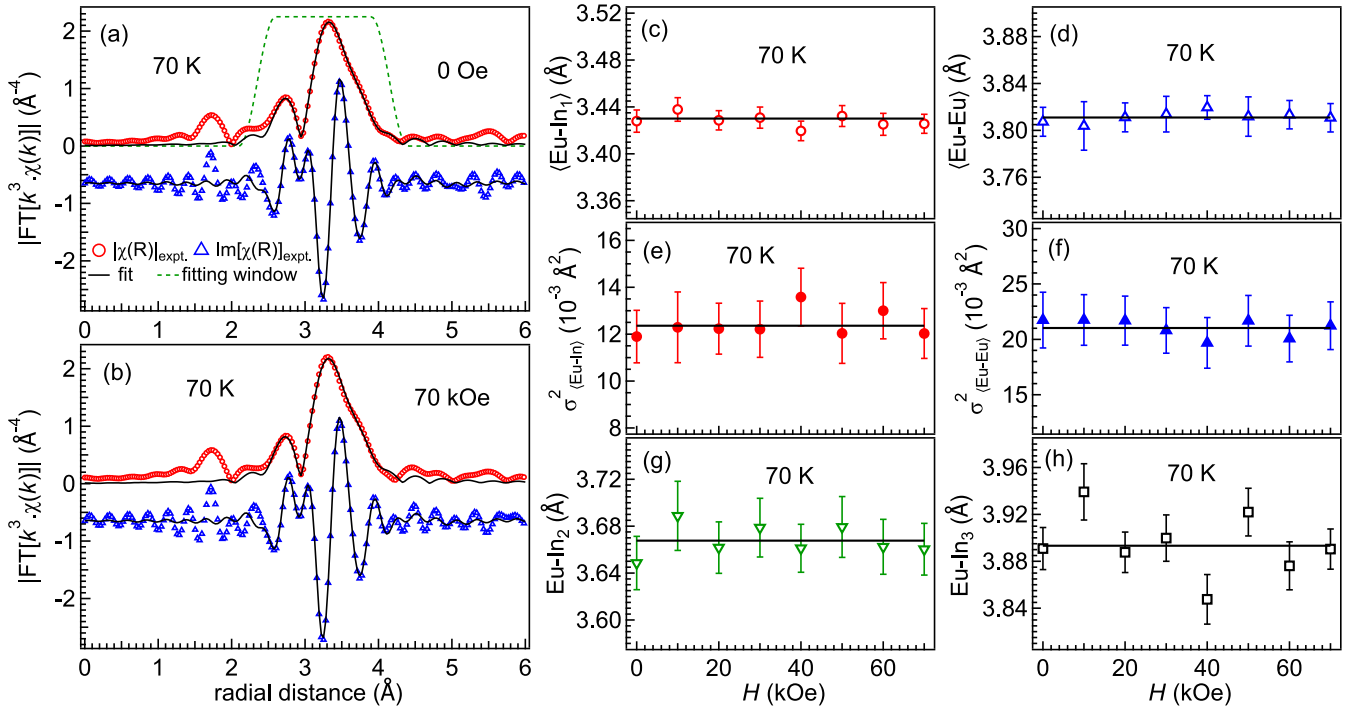


FIG. 8. (a), (b) Fits to the $|\chi(R)|$ and corresponding $\text{Im}[\chi(R)]$ spectra within the window indicated by the dotted green line [in panel (a)] at magnetic fields of 0 and 70 kOe, respectively, at 70 K. Each $\text{Im}[\chi(R)]$ spectrum is shifted downward for clarity. (c), (d) Magnetic-field dependence of the average Eu–In₁ and Eu–Eu scattering path distances, respectively. (e), (f) Corresponding Debye-Waller factors. (g), (h) Magnetic-field dependence of the Eu–In₂ and Eu–In₃ scattering path distances. Solid black lines indicate the field-independent behavior of the various scattering distances and their corresponding Debye-Waller factors.

(see Fig. S5 in [43] for the contributions from individual scattering paths). The magnetic-field dependence of the extracted parameters is summarized in Figs. 7(c)–7(h) and 8(c)–8(h). In all fits, S_0^2 was fixed at 0.793 and the resulting R factor (goodness of fit) lies in the range 0.004–0.006 at both temperatures.

Interestingly, the extracted Eu–In and Eu–Eu bond lengths, as well as their corresponding DW factors, exhibit no statistically significant field dependence up to 70 kOe at both 50 K (below T_c) and 70 K (above T_c). Since the DW factor captures contributions from both thermal and static disorder, the absence of any field-induced variation indicates that the applied magnetic field does not introduce additional local disorder nor does it drive a measurable local distortion (e.g., magnetostriction) in the Eu coordination environment. While extremely subtle structural changes below the sensitivity of the present EXAFS measurements (≈ 0.01 – 0.02 Å) cannot be completely excluded, our results suggest that the second anomaly observed in the specific heat is unlikely to be associated with a magnetic-field-driven local structural modification in Eu₂In.

IV. DISCUSSION

The change in the behavior of magnetocaloric properties of Eu₂In, namely, deviation of the $\Delta S_M^{\max}(H)$ curve, systematic variation of the local power exponent n , and evolution of the UMC around 25 kOe, points toward a modification in the nature of the transition at this critical field. At the same time, the occurrence of an unphysically small value of the power exponent n , the overshoot of the local exponent above

$n = 2$, and the persistence of dispersion in the UMC even for $H = 25$ – 70 kOe indicate that the overall transition retains its first-order character. It is important to note that ΔS_M is a cumulative quantity and its values at higher fields necessarily incorporate contributions from the low-field region [see Eq. (1)]. As a result, even if the transition evolves toward second-order behavior for $H > 25$ kOe, the imprint of the low-field first-order character would persist in the ΔS_M curves at higher fields, giving rise to first-order-like features in all derived quantities.

To test this scenario, we apply the Banerjee criterion, which states that negative and positive slopes of Arrott plots (H/M versus M^2) in the high-field region correspond to first-order and second-order transitions, respectively [55,56]. Figures 9(a)–9(e) display the modified Arrott plots (MAPs) [$M^{1/\beta}$ versus $(H/M)^{1/\gamma}$] for Eu₂In, constructed for different universality classes using their corresponding critical exponents (β and γ) [57]. If the transition indeed crossed over to second order at high fields, the MAPs would approach straight lines in the high-field regime for the correct choice of (β , γ) [58–60]. In contrast, none of the tested universality classes yield linear behavior in the high-field region. The insets of Figs. 9(a)–9(e), highlighting the 50–70 kOe field range, reveal a pronounced curvature that persists up to the highest fields investigated, confirming the robustness of the first-order character. Moreover, the slope of the MAPs in this regime changes from positive to negative with increasing temperature. To quantify this behavior, linear fits were performed in the 60–70 kOe range (see Figs. S6(a)–S6(e) of Ref. [43]) and the normalized slope is plotted in Fig. 9(f). A distinct crossover

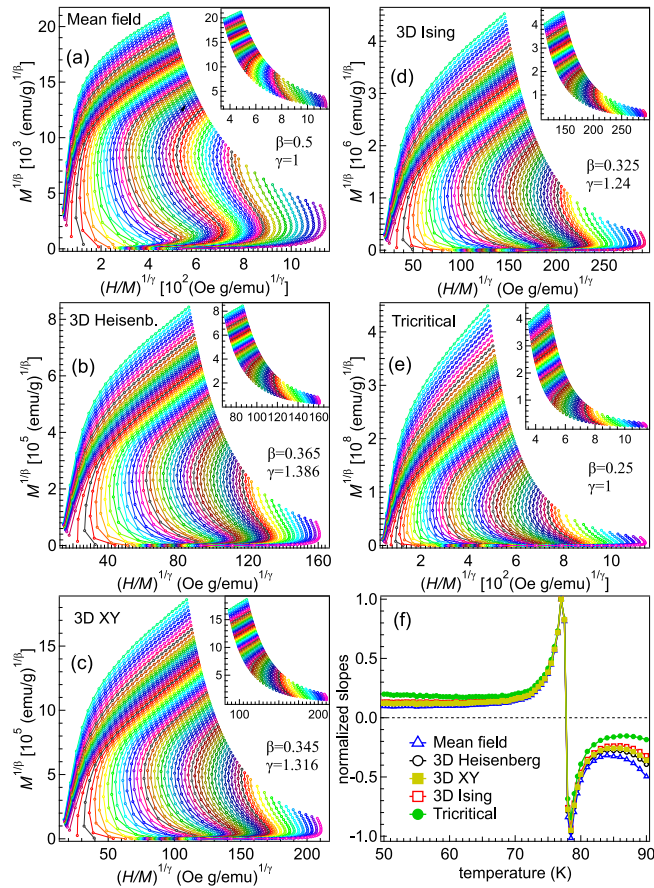


FIG. 9. (a)–(e) Modified Arrott plots (MAPs) [$M^{1/\beta}$ vs $(H/M)^{1/\gamma}$] of Eu_2In in the temperature range 40–90 K for different universality classes. Insets show an enlarged view of the high-field (50–70 kOe) region. (f) Normalized slopes of the MAPs obtained from 60 to 70 kOe field range for different universality classes.

from positive to negative slope occurs around 78 K for all universality classes, consistent with first-order-like behavior in which the critical exponents diverge at T_c . These results unambiguously demonstrate that the first-order nature of the transition in Eu_2In remains intact at least up to 70 kOe.

The splitting of the specific-heat peak under an external magnetic field, together with its correlation with the transition temperature observed in magnetization, and the field-dependent variation in the magnitude of the two peaks, suggest that a field-induced subtle change in the crystal structure of Eu_2In could be responsible for the LT peak in the C_p curves. However, at the same time, magnetic-field-dependent EXAFS measurements rule out any detectable field-induced local structural changes in the sample within the experimental resolution. Recently, Hardy *et al.* proposed a mathematical model describing a “two-step” process in the first-order transformation of giant magnetocaloric materials [15]. In this framework, the transformation initiates and terminates through abrupt changes in the transformed fraction, accompanied by large exchanges of latent heat [15]. In the present case, the external field may induce a similar two-step-like process in Eu_2In . Here it is important to mention that the field-induced splitting in the $C_p(T)$ curves of Eu_2In is fully

reversible, where the doublet reverts back to a singlet upon removal of the external field. We note that different sample geometries were used in the present measurements, namely an irregular bulk piece for magnetization, a thin rectangular plate for specific-heat measurements, and a powdered sample for XAS. Consequently, demagnetization effects, which lead to differences between the applied and internal magnetic fields, may vary among the different measurement techniques, particularly in the low-field regime. However, this effect is not expected to significantly influence the overall field dependence or the main conclusions of this study. A related two-stage transition has also been reported in Nd_2In , where the longitudinal strain first increases with magnetic field and then decreases at higher fields during its first-order transition at 108 K [22]. The authors correlate this two-stage transition in Nd_2In with theoretical calculations for Eu_2In , where the free energy with respect to FM order exhibits two minima [22,31]. At this stage, we believe that more comprehensive investigations such as magnetic-field-dependent diffraction [7,61] and/or a simultaneous probes of magnetic and structural responses [62–64] are required to clarify the origin of the two closely spaced peaks observed in the $C_p(T, H)$ curves of Eu_2In .

V. CONCLUSION

In summary, we have investigated the temperature- and magnetic-field-dependent magnetization, specific heat, and local crystal structure of Eu_2In across its first-order ferromagnetic–paramagnetic transition. The field dependence of ΔS_M^{max} follows a power-law behavior for $H \gtrsim 25$ kOe, but deviates at lower fields. The local field exponent n increases near T_c up to ~ 25 kOe and then decreases with further increase in field, which, together with the reduced dispersion in the universal master curve of $\Delta S_M(H, T)$ around 25 kOe, suggests a tendency of the transition to evolve toward second-order character above this critical field. Nevertheless, quantitative analyses of the magnetocaloric properties and modified Arrott plots unambiguously confirm that the overall first-order nature of the transition persists at least up to 70 kOe. While this behavior contrasts with theoretical predictions of a field-induced first- to second-order transition above ~ 25 kOe [32], it certainly indicates a subtle modification of the transition mechanism at higher fields. Specific-heat measurements reveal a clear field-induced splitting of the sharp, δ -like anomaly into a doublet. A comparison between the C_p peak positions and the T_c obtained from magnetization, together with the field-dependent changes in their relative amplitudes, suggests a nonmagnetic origin for the low-temperature peak. However, at the same time, EXAFS measurements show no detectable changes in the local crystal structure, ruling out a field-induced structural distortion in the sample within the experimental resolution. Taken together, these results point to a field-driven transformation of the transition in Eu_2In from a one-step to a two-step process.

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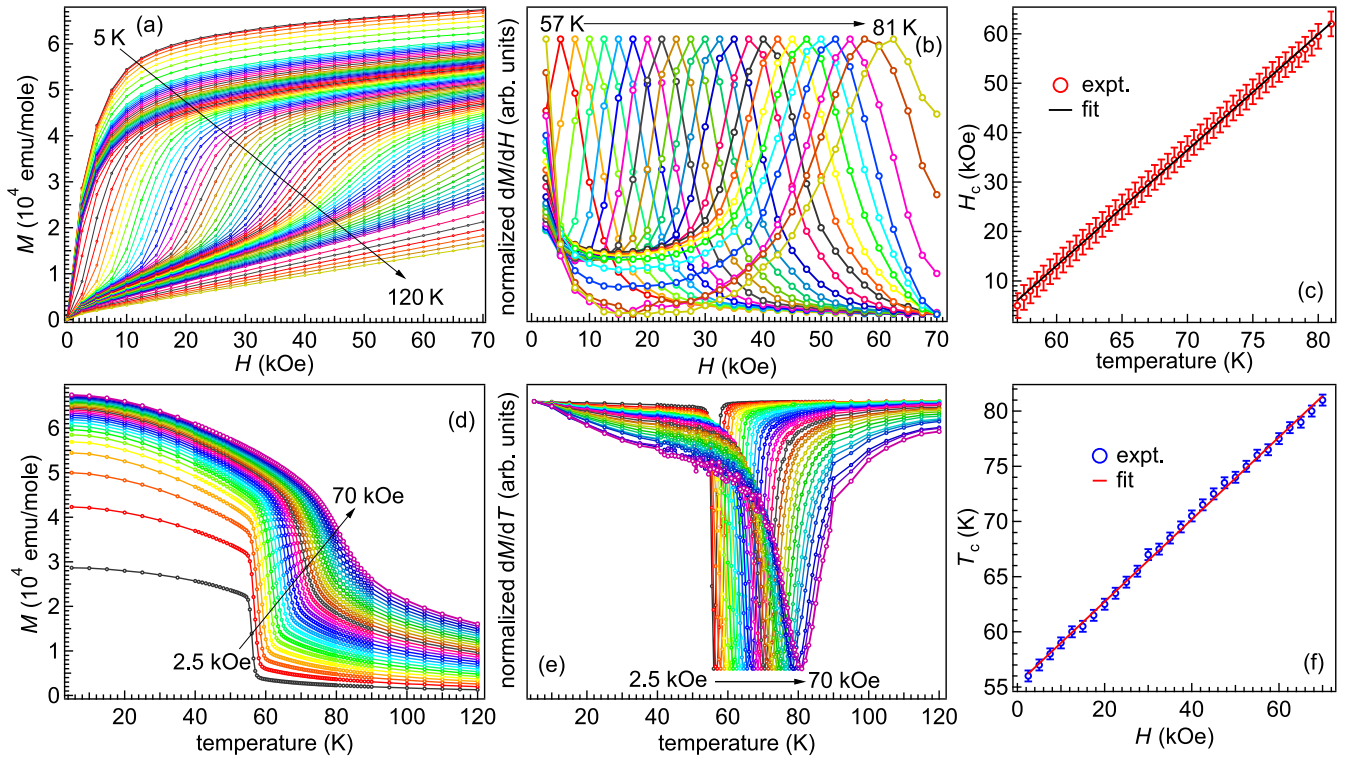


FIG. 10. (a) Virgin magnetization isotherms of Eu_2In measured from 0 to 70 kOe at different temperatures across T_c . (b) Normalized field derivative of the magnetization in the temperature range 57–81 K. (c) Temperature dependence of the critical field H_c ; the black solid line represents a linear fit to the data. (d) Temperature-dependent magnetization at different applied fields, extracted from the virgin magnetization isotherms. (e) Normalized temperature derivative of the magnetization at different fields. (f) Field dependence of the transition temperature; the red solid line represents a linear fit to the data.

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DATA AVAILABILITY

The data that support the findings of this article are openly available in the Harvard Dataverse repository at Ref. [65].

APPENDIX: ADDITIONAL EXPERIMENTAL DATA

1. Magnetization data

Figure 10(a) shows the virgin magnetization isotherms of Eu_2In measured from 0 to 70 kOe at different temperatures across T_c . For $T \leq T_c$, the curves show the saturating behavior similar to that observed at 2 K [Fig. 1(d)]; however, for $T > T_c$, the magnetic moment increases almost linearly for the lower applied magnetic fields and then shows as a step-like jump with further increase in the magnetic field [see

Fig. 10(a)]. The critical magnetic field (H_c) for this jump in the magnetization, i.e., the magnetic field required to ferromagnetically align the spins above T_c , increases with the temperature. In order to clearly present this, we show the normalized field derivative of the magnetization in Fig. 10(b) for $T > T_c$, where a monotonic shift in the peak position can be clearly observed to the higher field with increasing sample temperature. In Fig. 10(c), we plot this peak position as a func-

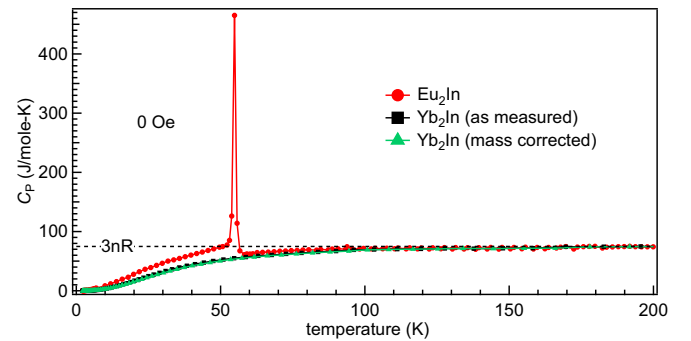


FIG. 11. Temperature-dependent specific heat of Eu_2In and its isostructural nonmagnetic reference compound Yb_2In measured in zero magnetic field using the 2τ relaxation method. The Yb_2In data are shown both as measured and after mass correction to account for the difference in atomic masses. The horizontal dashed line represents the classical Dulong-Petit limit.

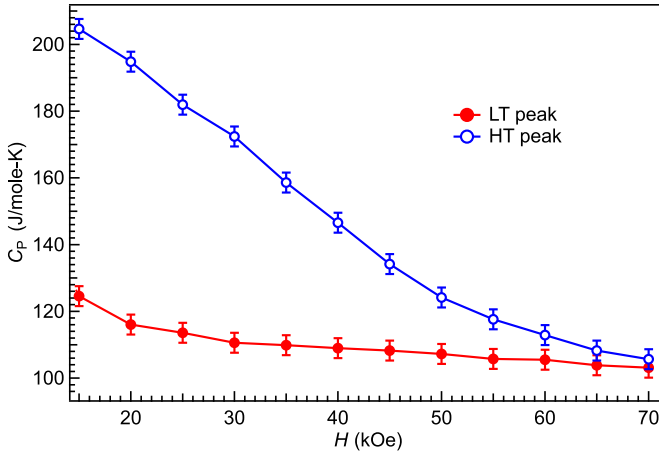


FIG. 12. Magnetic-field dependence of the peak values of the low- and high-temperature anomalies in the specific heat of Eu_2In .

tion of temperature, which shows that H_c increases linearly with temperature for $T > T_c$. Here, we perform the Gaussian fitting of the peaks in the vicinity of the transition to precisely estimate the peak position. The straight fit of the experimental data, as shown by the black solid line in Fig. 10(c), gives a significant shift of 2.35 kOe/K.

The temperature-dependent magnetization extracted from these virgin magnetization isotherms is presented in Fig. 10(d) at the different magnetic fields. With increase in the magnetic field, the transition becomes broader with a monotonic shift in the transition temperature to the higher values. The normalized dM/dT is plotted as a function of temperature in Fig. 10(e) at different fields to show the field evolution of the T_c . The field dependence of T_c is shown in Fig. 10(f), which

shows the linear behavior up to 70 kOe and a straight fit of the data gives a shift of 0.37 K/kOe in the transition temperature. This linear dependence of the transition temperature of Eu_2In on the external field makes it a suitable candidate for the device application in a wide range of the working temperature.

2. Specific heat data

Figure 11 shows the zero-field temperature-dependent specific heat data of Eu_2In (solid red circles). To account for the lattice (phonon) contribution to the specific heat of Eu_2In , we measured the $C_p(T)$ data of its isostructural nonmagnetic reference compound Yb_2In [18]. Since Eu_2In and Yb_2In have different formula unit masses, a correction for the mass difference was considered. Within the Debye approximation, the phonon heat capacity depends on the reduced temperature T/θ_D [see Eq. (5)]. The Debye temperature θ_D depends on the formula unit mass M as $\theta_D \propto M^{-1/2}$ [66]. Consequently, the measured temperature scale of Yb_2In was rescaled according to the relation [66]

$$T^* = T \left(\frac{M_{\text{Yb}_2\text{In}}}{M_{\text{Eu}_2\text{In}}} \right)^{1/2}. \quad (7)$$

The as-measured and mass-corrected $C_p(T)$ data of Yb_2In are shown in Fig. 11. No significant difference between the two data sets is observed in the present case. Therefore, the as-measured $C_p(T)$ data of Yb_2In are used to account for the lattice heat capacity of Eu_2In in all subsequent analyses [46,67].

Figure 12 shows the magnetic-field dependence of the peak magnitudes of the two anomalies observed in the specific-heat data of Eu_2In (see Fig. 4), after deconvolution for $H \geq 15$ kOe. The low-temperature (LT) peak is almost field invariant, whereas the high-temperature (HT) peak decreases monoton-

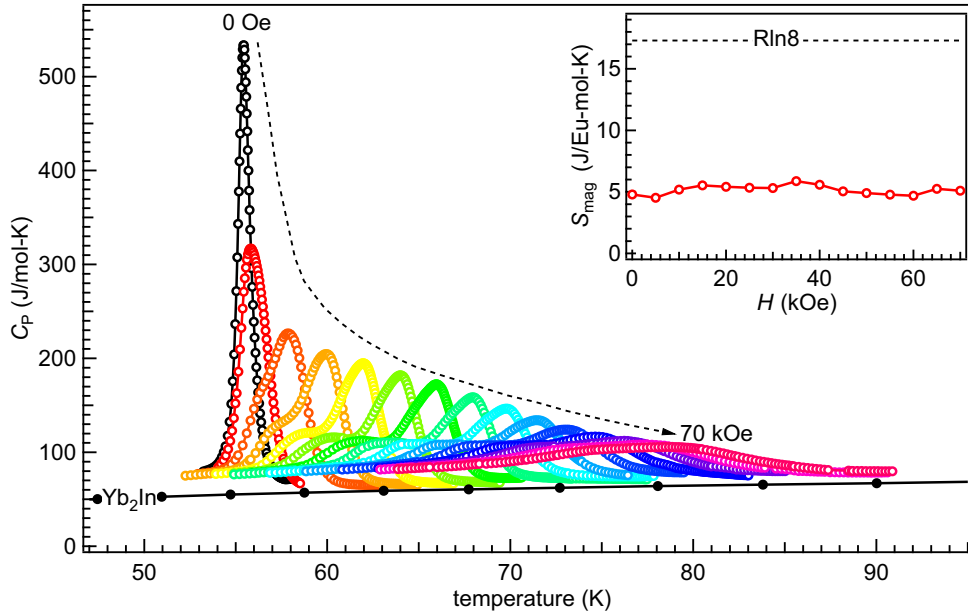


FIG. 13. Temperature-dependent specific heat $C_p(T)$ of Eu_2In in the vicinity of the magnetic transition measured using the long heat-pulse method at different applied magnetic fields. The black solid circles represent the $C_p(T)$ data of the nonmagnetic reference compound Yb_2In . Inset shows the magnetic entropy S_{mag} associated with the transition as a function of magnetic field. The horizontal dashed line denotes the theoretical entropy $R \ln(2J + 1)$ expected for Eu^{2+} ($J = 7/2$).

ically with increasing field and approaches the LT value near $H \approx 70$ kOe. The field invariance of the LT anomaly suggests a nonmagnetic (likely structural) origin of this feature.

Figure 13 shows the temperature dependence of the specific heat of Eu_2In near the magnetic transition under different applied magnetic fields, measured using the long heat-pulse method in heating mode. The $C_p(T)$ data of the isostructural nonmagnetic reference compound Yb_2In , shown by the black solid curve, are used to estimate the lattice contribution to the specific heat of Eu_2In . The magnetic entropy calculated by considering this lattice contribution is shown in the inset of Fig. 13. It is observed that the magnetic entropy asso-

ciated with the transition is nearly field independent and is significantly lower than the theoretical value expected for Eu^{2+} , as indicated by the horizontal dashed line in the inset of Fig. 13. Notably, the zero-field magnetic entropy associated with the transition is close to that reported in Ref. [19] (~ 5.7 J/Eu mol K), where a step-function approximation was used to estimate the lattice background in the transition region, despite the significantly larger peak value of C_p observed in Ref. [19]. This suggests that the transition in the present case is broader, possibly reflecting differences in compositional homogeneity between the two samples [68].

- [1] H. Hou, S. Qian, and I. Takeuchi, Materials, physics and systems for multicaloric cooling, *Nat. Rev. Mater.* **7**, 633 (2022).
- [2] R. Kainuma, Y. Imano, W. Ito, Y. Sutou, M. Morito, S. Okamoto, O. Kitakami, and T. Kanomata, Magnetic-field-induced shape recovery by reverse phase transformation, *Nature (London)* **439**, 957 (2006).
- [3] J. Mira, F. Rivadulla, J. Rivas, A. Fondado, T. Guidi, R. Caciuffo, F. Carsughi, P. G. Radaelli, and J. B. Goodenough, Structural transformation induced by magnetic field and “colossal-like” magnetoresistance response above 313 K in MnAs , *Phys. Rev. Lett.* **90**, 097203 (2003).
- [4] J. G. Roch, D. Miserev, G. Froehlicher, N. Leisgang, L. Sponfeldner, K. Watanabe, T. Taniguchi, J. Klinovaja, D. Loss, and R. J. Warburton, First-order magnetic phase transition of mobile electrons in monolayer MoS_2 , *Phys. Rev. Lett.* **124**, 187602 (2020).
- [5] V. K. Pecharsky and K. A. Gschneidner Jr., Giant magnetocaloric effect in $\text{Gd}_5(\text{Si}_2\text{Ge}_2)$, *Phys. Rev. Lett.* **78**, 4494 (1997).
- [6] N. A. de Oliveira, P. J. von Ranke, M. V. T. Costa, and A. Troper, Magnetocaloric effect in the intermetallic compounds RCO_2 ($\text{R} = \text{Dy}, \text{Ho}, \text{Er}$), *Phys. Rev. B* **66**, 094402 (2002).
- [7] V. K. Pecharsky, A. P. Holm, K. A. Gschneidner, and R. Rink, Massive magnetic-field-induced structural transformation in Gd_5Ge_4 and the nature of the giant magnetocaloric effect, *Phys. Rev. Lett.* **91**, 197204 (2003).
- [8] A. Kumar, A. Biswas, and Y. Mudryk, Stability of the first-order character of phase transition in HoCo_2 , *J. Appl. Phys.* **136**, 203902 (2024).
- [9] W. Choe, V. K. Pecharsky, A. O. Pecharsky, K. A. Gschneidner, V. G. Young, and G. J. Miller, Making and breaking covalent bonds across the magnetic transition in the giant magnetocaloric material $\text{Gd}_5(\text{Si}_2\text{Ge}_2)$, *Phys. Rev. Lett.* **84**, 4617 (2000).
- [10] Y. Mozharivskyj, A. O. Pecharsky, V. K. Pecharsky, and G. J. Miller, On the high-temperature phase transition of $\text{Gd}_5\text{Si}_2\text{Ge}_2$, *J. Am. Chem. Soc.* **127**, 317 (2005).
- [11] M. Piazza, C. Bennati, and V. Basso, Thermodynamics of the heat-flux avalanches at the first-order magnetic transition in magnetocaloric materials, *Phys. Rev. Appl.* **8**, 044023 (2017).
- [12] F. Erbesdobler, C. R. H. Bahl, R. Bjørk, and K. K. Nielsen, Detailed isofield calorimetry of $\text{La}(\text{Fe}, \text{Si}, \text{Mn})\text{H}$ reveals distributed magnetocaloric phase transitions, *J. Appl. Phys.* **127**, 095102 (2020).
- [13] C. Bennati, L. Gozzelino, E. S. Olivetti, and V. Basso, Heterogeneous nucleation and heat flux avalanches in $\text{La}(\text{Fe}, \text{Si})_{13}$ magnetocaloric compounds near the critical point, *Appl. Phys. Lett.* **109**, 231904 (2016).
- [14] F. Guillou, H. Yibole, Z. Q. Ou, E. Brück, and O. Tegus, Large recalescence-like event at the first cooling across the magnetic transition of $(\text{Mn}, \text{Fe})_2(\text{P}, \text{Si})$ magnetocaloric materials, *Scr. Mater.* **160**, 81 (2019).
- [15] V. Hardy, R. Hamane, F. Veillon, M. Risser, and F. Guillou, “Two-steps” process in the first-order transformation of giant magnetocaloric materials, *Acta Mater.* **231**, 117869 (2022).
- [16] P. J. Shamberger and F. S. Ohuchi, Hysteresis of the martensitic phase transition in magnetocaloric-effect Ni-Mn-Sn alloys, *Phys. Rev. B* **79**, 144407 (2009).
- [17] V. Provenzano, A. J. Shapiro, and R. D. Shull, Reduction of hysteresis losses in the magnetic refrigerant $\text{Gd}_5\text{Ge}_2\text{Si}_2$ by the addition of iron, *Nature (London)* **429**, 853 (2004).
- [18] A. Biswas, A. Kumar, P. Singh, and Y. Mudryk, From complex magnetic ground states to magnetocaloric effects: A review of rare earth R_2In intermetallic compounds, *J. Phys. D: Appl. Phys.* **59**, 053001 (2026).
- [19] F. Guillou, A. K. Pathak, D. Paudyal, Y. Mudryk, F. Wilhelm, A. Rogalev, and V. K. Pecharsky, Non-hysteretic first-order phase transition with large latent heat and giant low-field magnetocaloric effect, *Nat. Commun.* **9**, 2925 (2018).
- [20] A. Biswas, N. A. Zarkevich, A. K. Pathak, O. Dolotko, I. Z. Hlova, A. V. Smirnov, Y. Mudryk, D. D. Johnson, and V. K. Pecharsky, First-order magnetic phase transition in Pr_2In with negligible thermomagnetic hysteresis, *Phys. Rev. B* **101**, 224402 (2020).
- [21] A. Biswas, R. K. Chouhan, A. Thayer, Y. Mudryk, I. Z. Hlova, O. Dolotko, and V. K. Pecharsky, Unusual first-order magnetic phase transition and large magnetocaloric effect in Nd_2In , *Phys. Rev. Mater.* **6**, 114406 (2022).
- [22] W. Liu, F. Scheibel, T. Gottschall, E. Bykov, I. Dirba, K. Skokov, and O. Gutfleisch, Large magnetic entropy change in Nd_2In near the boiling temperature of natural gas, *Appl. Phys. Lett.* **119**, 022408 (2021).
- [23] B. Gegen, B. Huhe, Z.-Q. Ou, F. Guillou, and H. Yibole, Thermal hysteresis and reversibility of the giant magnetocaloric effect at the ferromagnetic transition of Nd_2In , *Materials* **18**, 3104 (2025).

- [24] W. Cui, G. Yao, S. Sun, Q. Wang, J. Zhu, and S. Yang, Unconventional metamagnetic phase transition in $R_2\text{In}$ ($R = \text{Nd, Pr}$) with lambda-like specific heat and nonhysteresis, *J. Mater. Sci. Technol.* **101**, 80 (2022).
- [25] M. Forker, R. Müßeler, S. C. Bedi, M. Olzon-Dionysio, and S. D. de Souza, Magnetic and electric hyperfine interactions in the rare-earth indium compounds $R_2\text{In}$ studied by ^{111}Cd perturbed angular correlations, *Phys. Rev. B* **71**, 094404 (2005).
- [26] H. Gamari-Seale, T. Anagnostopoulos, and J. K. Yakinthos, Magnetic characteristics of rare-earth indium $R_2\text{In}$ ($R = \text{Y, Nd, Sm, Gd, Tb, Dy, Ho, Er, and Tm}$) intermetallic compounds, *J. Appl. Phys.* **50**, 434 (1979).
- [27] F. Guillou, H. Yibole, R. Hamane, V. Hardy, Y. B. Sun, J. J. Zhao, Y. Mudryk, and V. K. Pecharsky, Crystal structure and physical properties of Yb_2In and $\text{Eu}_{2-x}\text{Yb}_x\text{In}$ alloys, *Phys. Rev. Mater.* **4**, 104402 (2020).
- [28] W. Baela and A. Szytuła, Crystal structure and magnetic properties of RE_2In compounds, *J. Less-Common Met.* **138**, 123 (1988).
- [29] C. Ritter, D. H. Ryan, A. Provino, G. Lamura, Y. Mudryk, R. K. Chouhan, P. Singh, D. D. Johnson, V. K. Pecharsky, and P. Manfrinetti, Magnetic ordering in Eu_2In and Eu_2Sn , *J. Alloys Compd.* **980**, 173573 (2024).
- [30] D. H. Ryan, D. Paudyal, F. Guillou, Y. Mudryk, A. K. Pathak, and V. K. Pecharsky, The first-order magnetoelastic transition in Eu_2In : A ^{151}Eu Mössbauer study, *AIP Adv.* **9**, 125137 (2019).
- [31] E. Mendive-Tapia, D. Paudyal, L. Petit, and J. B. Staunton, First-order ferromagnetic transitions of lanthanide local moments in divalent compounds: An itinerant electron positive feedback mechanism and Fermi surface topological change, *Phys. Rev. B* **101**, 174437 (2020).
- [32] B. P. Alho, P. O. Ribeiro, P. J. von Ranke, F. Guillou, Y. Mudryk, and V. K. Pecharsky, Free-energy analysis of the nonhysteretic first-order phase transition of Eu_2In , *Phys. Rev. B* **102**, 134425 (2020).
- [33] A. P. Holm, V. K. Pecharsky, K. A. Gschneidner Jr., R. Rink, and M. N. Jirmanus, X-ray powder diffractometer for in situ structural studies in magnetic fields from 0 to 35 kOe between 2.2 and 315 K, *Rev. Sci. Instrum.* **75**, 1081 (2004).
- [34] V. Hardy, Y. Bréard, and C. Martin, Derivation of the heat capacity anomaly at a first-order transition by using a semi-adiabatic relaxation technique, *J. Phys.: Condens. Matter* **21**, 075403 (2009).
- [35] J. C. Lashley, M. F. Hundley, A. Migliori, J. L. Sarrao, P. G. Pagliuso, T. W. Darling, M. Jaime, J. C. Cooley, W. L. Hults, L. Morales, D. J. Thoma, J. L. Smith, J. Boerio-Goates, B. F. Woodfield, G. R. Stewart, R. A. Fisher, and N. E. Phillips, Critical examination of heat capacity measurements made on a quantum design physical property measurement system, *Cryogenics* **43**, 369 (2003).
- [36] G. Bunker, *Introduction to XAFS: A Practical Guide to X-ray Absorption Fine Structure Spectroscopy* (Cambridge University Press, Cambridge, UK, 2010).
- [37] A. Kumar, R. Shukla, R. Kumar, R. J. Choudhary, S. N. Jha, and R. S. Dhaka, Probing the electronic and local structure of $\text{Sr}_{2-x}\text{La}_x\text{CoNbO}_6$ using near-edge and extended x-ray absorption fine structures, *Phys. Rev. B* **105**, 245155 (2022).
- [38] B. Ravel and M. Newville, *ATHENA, ARTEMIS, HEPHAESTUS*: Data analysis for x-ray absorption spectroscopy using *IFEFFIT*, *J. Synchrotron Rad.* **12**, 537 (2005).
- [39] S. I. Zabinsky, J. J. Rehr, A. Ankudinov, R. C. Albers and M. J. Eller, Multiple-scattering calculations of x-ray-absorption spectra, *Phys. Rev. B* **52**, 2995 (1995).
- [40] M. H. Phan and S. C. Yu, Review of magnetocaloric effect in manganite materials, *J. Magn. Magn. Mater.* **308**, 325 (2007).
- [41] A. Kumar and R. S. Dhaka, Unraveling magnetic interactions and the spin state in insulating $\text{Sr}_{2-x}\text{La}_x\text{CoNbO}_6$, *Phys. Rev. B* **101**, 094434 (2020).
- [42] V. Franco, J. S. Blázquez, and A. Conde, Field dependence of the magnetocaloric effect in materials with a second order phase transition: A master curve for the magnetic entropy change, *Appl. Phys. Lett.* **89**, 222512 (2006).
- [43] See Supplemental Material at <http://link.aps.org/supplemental/10.1103/htcd-q2t7> for the details of the magnetization, specific heat, and XAS data.
- [44] J. Y. Law, V. Franco, L. M. Moreno-Ramírez, A. Conde, D. Y. Karpenkov, I. Radulov, K. P. Skokov, and O. Gutfleisch, A quantitative criterion for determining the order of magnetic phase transitions using the magnetocaloric effect, *Nat. Commun.* **9**, 2680 (2018).
- [45] C. M. Bonilla, J. Herrero-Albillos, F. Bartolomé, L. M. García, M. Parra-Borderías, and V. Franco, Universal behavior for magnetic entropy change in magnetocaloric materials: An analysis on the nature of phase transitions, *Phys. Rev. B* **81**, 224424 (2010).
- [46] V. K. Anand, D. T. Adroja, and A. D. Hillier, Ferromagnetic cluster spin-glass behavior in PrRhSn_3 , *Phys. Rev. B* **85**, 014418 (2012).
- [47] C. Kittel, *Introduction to Solid State Physics*, 8th ed. (Wiley, New York, 2005).
- [48] A. Kumar, A. Biswas, and Y. Mudryk, Magnetic glassiness in noncentrosymmetric Sm_7Pd_3 : Interplay of magnetic frustration, long-range order, and frozen domains, *Phys. Rev. B* **111**, 174448 (2025).
- [49] W. Liu, F. Scheibel, N. Fortunato, I. Dirba, T. Gottschall, H. Zhang, K. Skokov, and O. Gutfleisch, Role of Debye temperature in achieving large adiabatic temperature changes at cryogenic temperatures: A case study on Pr_2In , *Phys. Rev. B* **109**, L140407 (2024).
- [50] V. K. Pecharsky, K. A. Gschneidner, Jr., Y. Mudryk, and D. Paudyal, Making the most of the magnetic and lattice entropy changes, *J. Magn. Magn. Mater.* **321**, 3541 (2009).
- [51] A. Kumar, P. Singh, A. Doyle, D. L. Schlagel, and Y. Mudryk, Multiple magnetic interactions and large inverse magnetocaloric effect in TbSi and $\text{TbSi}_{0.6}\text{Ge}_{0.4}$, *Phys. Rev. B* **109**, 214410 (2024).
- [52] W. B. Jiang, M. Smidman, W. Xie, J. Y. Liu, J. M. Lee, J. M. Chen, S. C. Ho, H. Ishii, K. D. Tsuei, C. Y. Guo, Y. J. Zhang, H. Lee, and H. Q. Yuan, Antiferromagnetism with divalent Eu in EuNi_5As_3 , *Phys. Rev. B* **95**, 024416 (2017).
- [53] N. M. Souza-Neto, J. Zhao, E. E. Alp, G. Shen, S. V. Sinogeikin, G. Lapertot, and D. Haskel, Reentrant valence transition in EuO at high pressures: Beyond the bond-valence model, *Phys. Rev. Lett.* **109**, 026403 (2012).
- [54] X. Tan, G. Fabbri, D. Haskel, A. A. Yaroslavtsev, H. Cao, C. M. Thompson, K. Kovnir, A. P. Menushenkov, R. V. Chernikov, V. O. Garlea, and M. Shatruk, A transition from localized to strongly correlated electron behavior and mixed valence driven by physical or chemical pressure in ACo_2As_2 ($A = \text{Eu and Ca}$), *J. Am. Chem. Soc.* **138**, 2724 (2016).

- [55] A. Arrott and J. Noakes, Approximate equation of state for nickel near its critical temperature, *Phys. Rev. Lett.* **19**, 786 (1967).
- [56] S. K. Banerjee, On a generalised approach to first and second order magnetic transitions, *Phys. Lett.* **12**, 16 (1964).
- [57] J. C. Le Guillou and J. Zinn-Justin, Critical exponents from field theory, *Phys. Rev. B* **21**, 3976 (1980).
- [58] P. Sarkar, P. Mandal, A. K. Bera, S. M. Yusuf, L. S. Sharath Chandra, and V. Ganesan, Field-induced first-order to second-order magnetic phase transition in $\text{Sm}_{0.52}\text{Sr}_{0.48}\text{MnO}_3$, *Phys. Rev. B* **78**, 012415 (2008).
- [59] S. Rößler, U. K. Rößler, K. Nenkov, D. Eckert, S. M. Yusuf, K. Dörr, and K.-H. Müller, Rounding of a first-order magnetic phase transition in Ga-doped $\text{La}_{0.67}\text{Ca}_{0.33}\text{MnO}_3$, *Phys. Rev. B* **70**, 104417 (2004).
- [60] A. Kumar, A. Biswas, and Y. Mudryk, Nontrivial critical behavior at magnetic transitions: A case study of Sm_7Pd_3 , *Phys. Rev. B* **112**, 224416 (2025).
- [61] Y. Mudryk, D. Paudyal, V. K. Pecharsky, K. A. Gschneidner, S. Misra, and G. J. Miller, Controlling magnetism of a complex metallic system using atomic individualism, *Phys. Rev. Lett.* **105**, 066401 (2010).
- [62] A. Aubert, K. Skokov, G. Gomez, A. Chirkova, I. Radulov, F. Wilhelm, A. Rogalev, H. Wende, O. Gutfleisch, and K. Ollefs, Simultaneous multi-property probing during magneto-structural phase transitions: An element-specific and macroscopic hysteresis characterization at ID12 of the ESRF, *IEEE Trans. Instrum. Meas.* **71**, 1 (2022).
- [63] K. P. Skokov, A. Y. Karpenkov, D. Y. Karpenkov, I. A. Radulov, D. Gunzing, B. Eggert, A. Rogalev, F. Wilhelm, J. Liu, Y. Shao, K. Ollefs, M. E. Gruner, H. Wende, and O. Gutfleisch, A multi-stage, first-order phase transition in $\text{LaFe}_{11.8}\text{Si}_{1.2}$: Interplay between the structural, magnetic, and electronic degrees of freedom, *Appl. Phys. Rev.* **10**, 031408 (2023).
- [64] D. Yu Karpenkov, A. Yu Karpenkov, K. P. Skokov, I. A. Radulov, M. Zheleznyi, T. Faske, and O. Gutfleisch, Pressure dependence of magnetic properties in $\text{La}(\text{Fe}, \text{Si})_{13}$: Multi-stimulus responsiveness of caloric effects by modeling and experiment, *Phys. Rev. Appl.* **13**, 034014 (2020).
- [65] A. Kumar, A. Biswas, T. A. Tyson, D. Haskel, C. J. Pollock, and Y. Mudryk, Replication Data for “Magnetic field induced modification of a first-order ferromagnetic transition in Eu_2In ”, Harvard Dataverse, V1. (2026), <https://doi.org/10.7910/DVN/JAX1JL>.
- [66] V. K. Anand and D. C. Johnston, Antiferromagnetism in EuCu_2As_2 and $\text{EuCu}_{1.82}\text{Sb}_2$ single crystals, *Phys. Rev. B* **91**, 184403 (2015).
- [67] V. K. Anand, Z. Hossain, G. Chen, M. Nicklas, and C. Geibel, Heavy fermion behavior in $\text{PrRh}_2\text{B}_2\text{C}$: Excitonic mass enhancement, *Phys. Rev. B* **79**, 113107 (2009).
- [68] N. J. Jones, H. Ucar, J. J. Ipus, M. E. McHenry, and D. E. Laughlin, The effect of distributed exchange parameters on magnetocaloric refrigeration capacity in amorphous and nanocomposite materials, *J. Appl. Phys.* **111**, 07A334 (2012).